



清华大学
Tsinghua University



MAE Thematic Seminar
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南方科技大学
SOUTHERN UNIVERSITY OF SCIENCE AND TECHNOLOGY

Mechanics and Tribology of Nano-/Microscale Solid Interfaces:

From “Zero Friction” to Force-Driven Chemistry and Nanofracture

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Mechanics and Tribology of Solid Interfaces

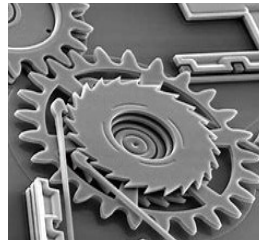
Tribology: the science and engineering of friction, adhesion, wear, and lubrication



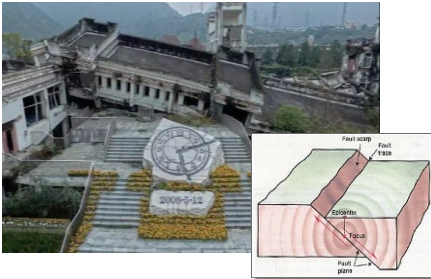
Energy & Environment



Machining & Processing



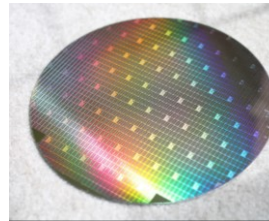
Microdevice



Natural disasters



Health care



Semiconductor manufacturing

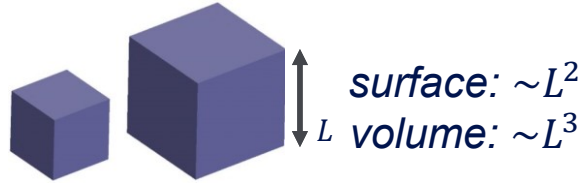
Economic impact:

- Friction consumes ~30% global primary energy;
- Wear causes ~80% machine failures;
- Economic loss: 2%~7% of GDP

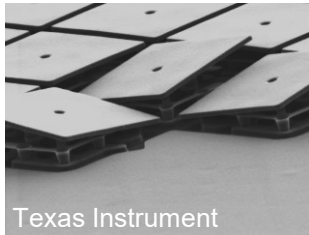
H. Liu, et al, *Friction* 11, 839 (2023).

Solid Interfaces on Nano-/Microscales

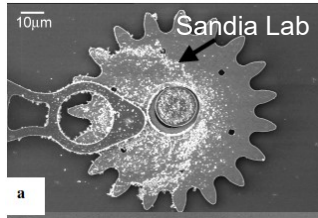
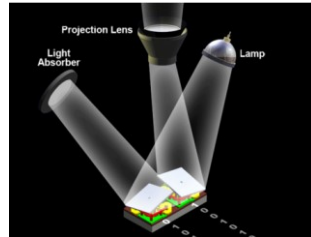
Why nano-/microscale?



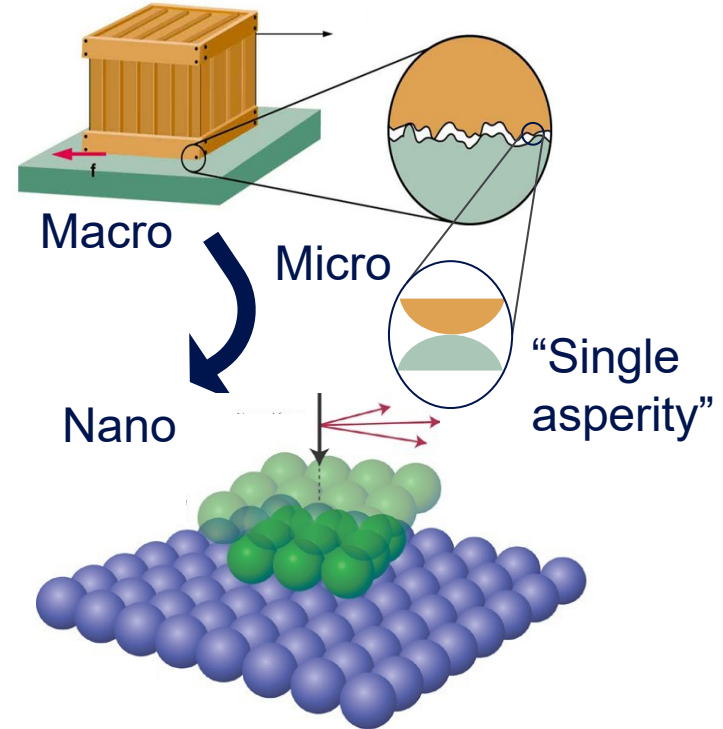
Surface-to-volume ratio



Digital micro-mirror arrays



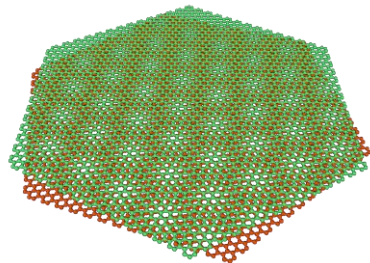
**Tribology in
microdevices**



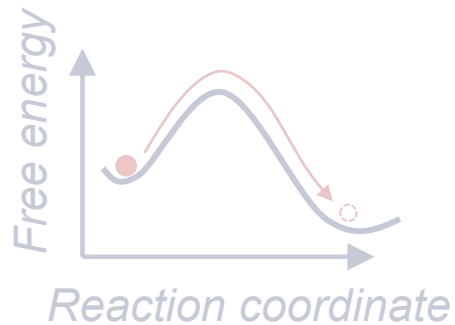
M. Urbakh & E. Meyer. *Nat. Mater.* 9, 8–10 (2010).

Outline

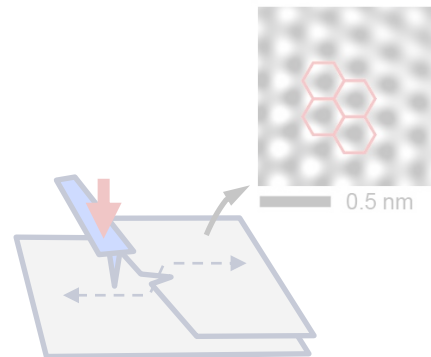
sliding



Structural
Superlubricity



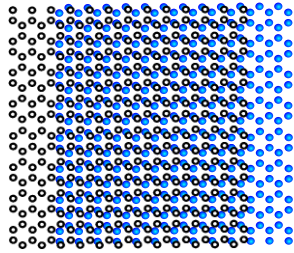
Mechano-
chemistry



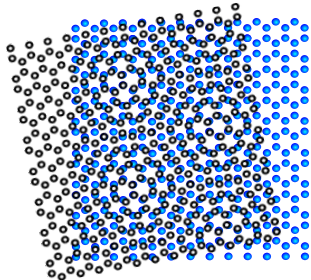
Nanofracture of
2D Materials

Structural Superlubricity: “Zero Friction” without Lubricants

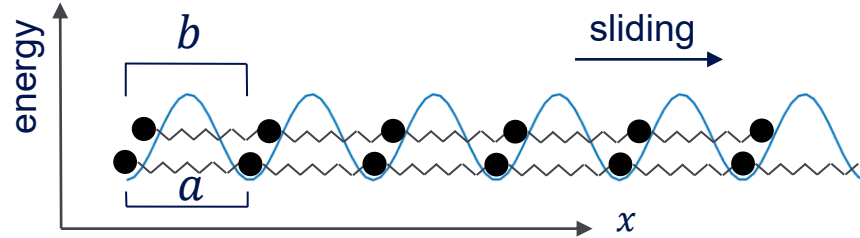
The atomic chain model:



*Commensurate:
high friction*



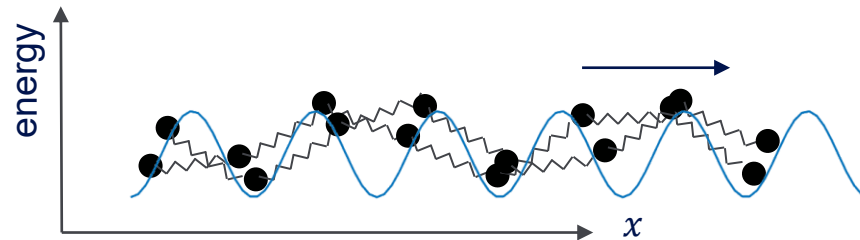
*Incommensurate:
superlubricity*



$$\frac{a}{b} = 1$$

Each atom: $U(x_i) = U_0 \sin\left(\frac{x_i}{2\pi a}\right)$

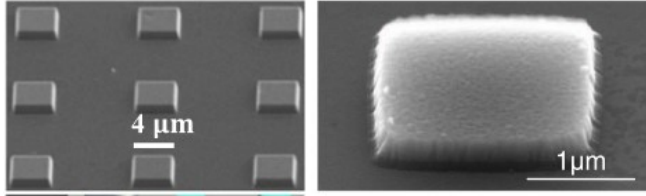
Total potential: $U_{\text{tot}}(x) = NU(x_i) \rightarrow \text{friction} \sim -\frac{dU_{\text{tot}}}{dx}$



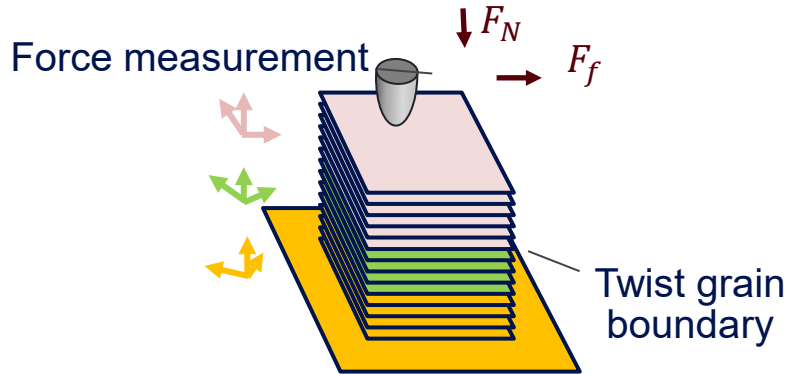
$$\frac{a}{b} = \sqrt{2}$$

U_{tot} is translational invariant: no friction

The “Graphite Mesa” System



Z. Liu, et al. *Phys. Rev. Lett.* 108, 205503 (2012).

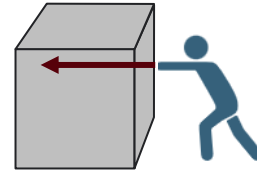


K. Wang, C. Qu*, J. Wang, B. Quan, and Q. Zheng*,
Phys. Rev. Lett. 125, 026101 (2020).

Friction measurement results:

- Superlubricity: ~ 0.01 MPa
- Shear strength (steel):
100~600 MPa

10 t weight, 1x1 m², 0.1% contact area

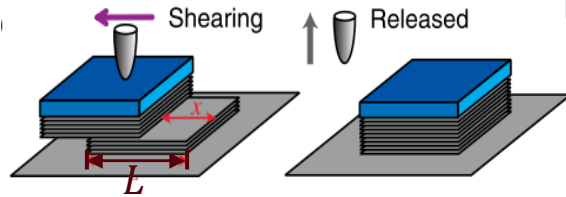


Force to move:

- Non-superlubric: 10 kN
- Superlubric: 1 N

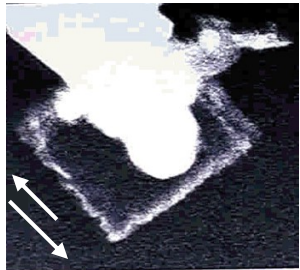
Unique Behaviors under Near-Zero Friction

The “self-retraction” motion:



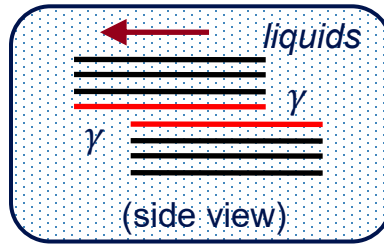
Retraction force:

$$F_{\text{ret}} = -\frac{\partial U}{\partial x} \\ \approx -2\gamma L$$



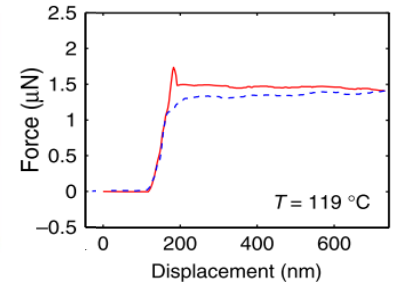
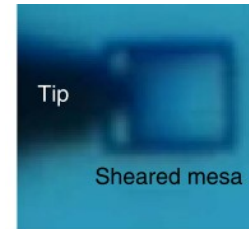
(top view)

Q. Zheng, et al, *Phys. Rev. Lett.* 100, 067205 (2008).



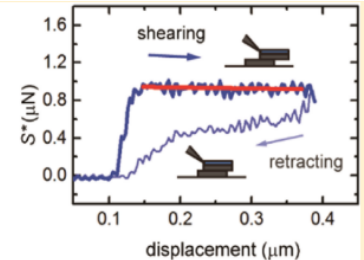
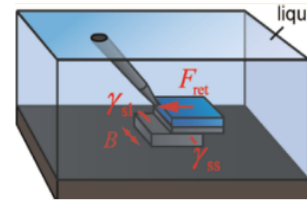
Surface energy: γ

measure **surface energy**



W. Wang, et al, *Nat. Commun.* 6, 7853 (2015).

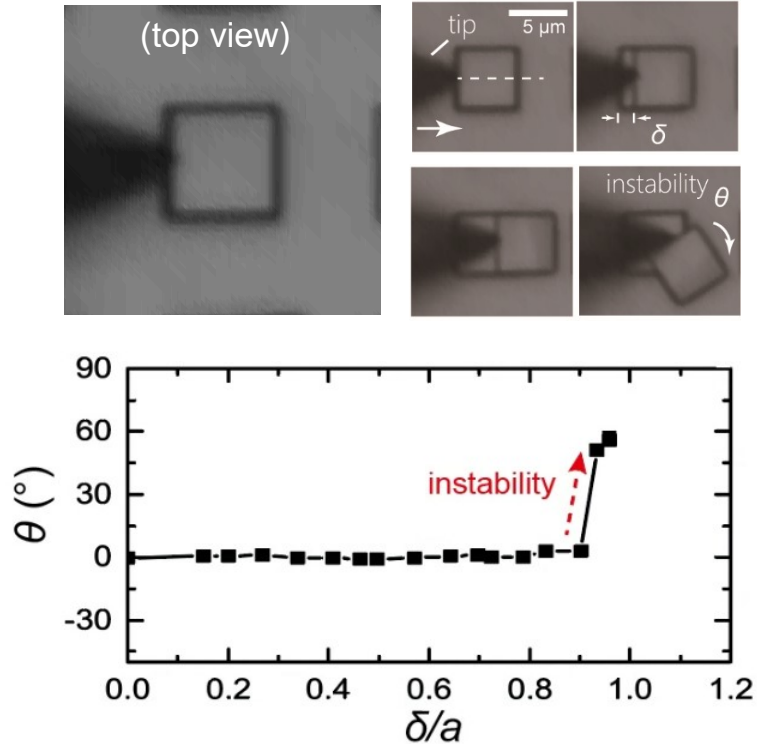
measure solid-liquid **interfacial energy**



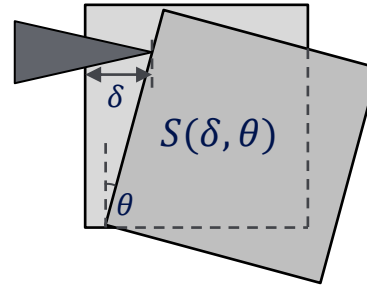
C. Qu, et al, *J. Phys. Chem. C* 123, 11671 (2019).

Unique Behaviors under Near-Zero Friction

The “rotational instability”:

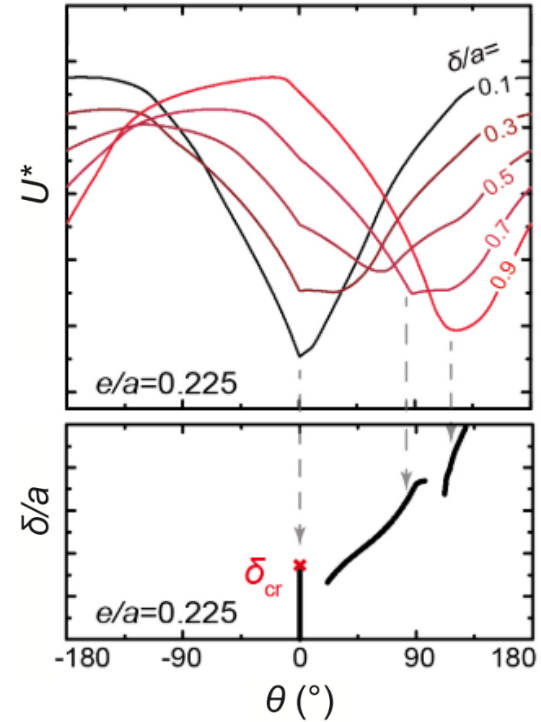


A simple mechanics model



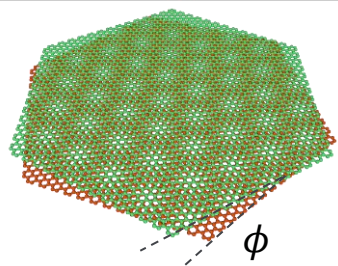
$$\underline{U}(\delta, \theta) \sim -2\underline{\gamma} \underline{S}(\delta, \theta)$$

total free energy overlapping area



Unique Behaviors under Near-Zero Friction

Controlling rotation is crucial:



Friction force:

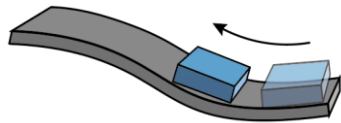
Commensurate

$\gg$

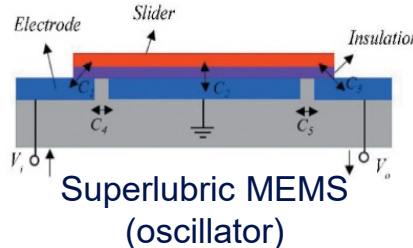
10,000x incommensurate

Maintaining the
misfit angle

Possible applications:



Superlubric "slider-track"



Superlubric MEMS
(oscillator)

C. Qu, et al. *J. Harbin Inst. Technol.* 27, 45 (2020)

Highlighted by Nature Materials:

MATERIAL WITNESS

A new twist on superlubricity

Lubrication has traditionally relied on fluids and other additives to reduce frictional dissipation as one solid surface moves over another. But over the past decade or so it has become clear that dry sliding at the microscale can become almost frictionless when there is incommensurability in the atomic-scale structure of the two surfaces¹. This effect was predicted in the 1980s², and dubbed superlubricity in 1991³ — but only more recently have technological developments in microscale fabrication and manipulation advanced to the point where it can be seen and investigated.

so as to minimize the interfacial energy — something only possible when the frictional forces are so low that this interfacial energy becomes the dominant determinant of the structure's configuration.

Quanshui Zheng of Tsinghua University — one of the authors of that earlier study — and co-workers have now uncovered a new aspect of this phenomenon. They find that a flake pushed in this manner may spontaneously rotate after it has moved a certain distance, in order to move into a lower-energy commensurate relationship with the lower surface⁴. Once the rotation

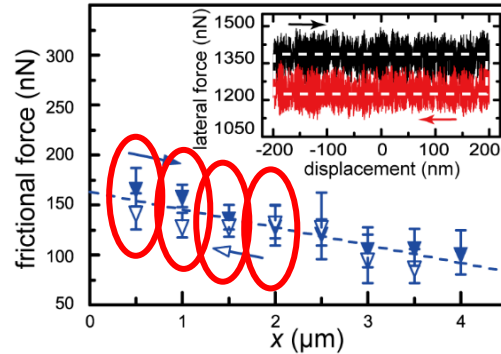
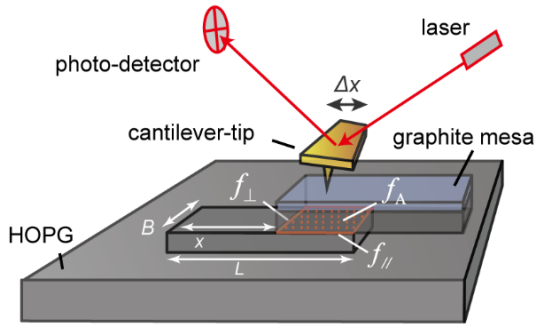
*" ... understanding this rotational effect in superlubricity should **enable it to be controlled**, ... for a rectangular flake sliding almost friction-free along a 'guide rail' ... the slider **literally stays on track.**"*

Nature Materials. 18 (8), 774-774 (2019).

Contact Edges in Superlubricity

Where does the non-zero friction come from?

Atomic force microscope (AFM)

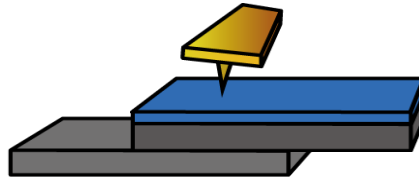


Contact area: $A = B(L - x)$

Contact edge length:

$$C = 2(L - x) + 2B$$

$$F_f = f_{\text{area}}A + f_{\text{edge}}C$$



Edge atom: 1.1 ± 0.1 pN/atom

Inner atom: -0.05 ± 0.05 fN/atom

≈ 0 in current noise level

Edge atoms:

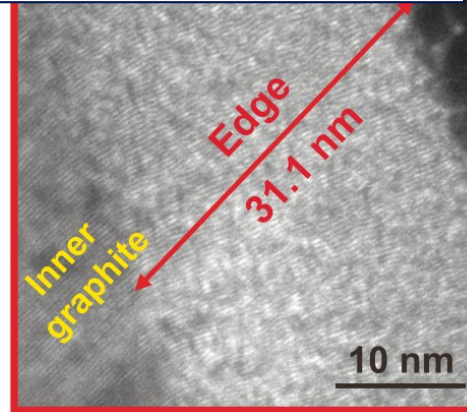
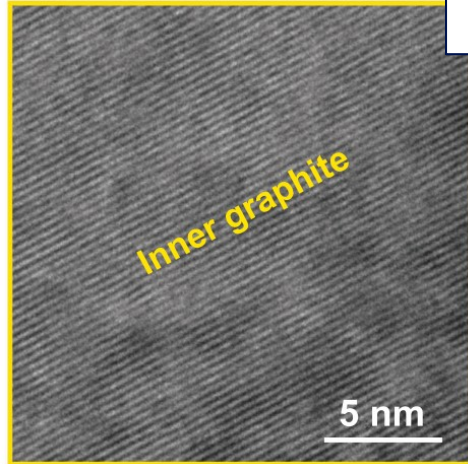
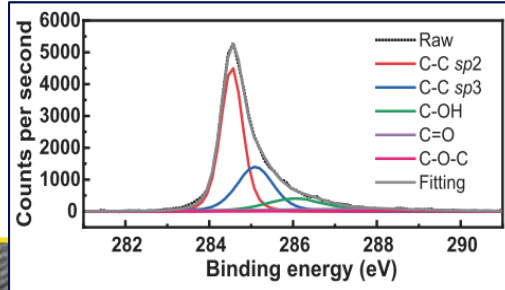
10,000x larger friction than inner atoms

1st experiment to decouple edge- and area-induced friction

Implications of Edge-Dominated Friction

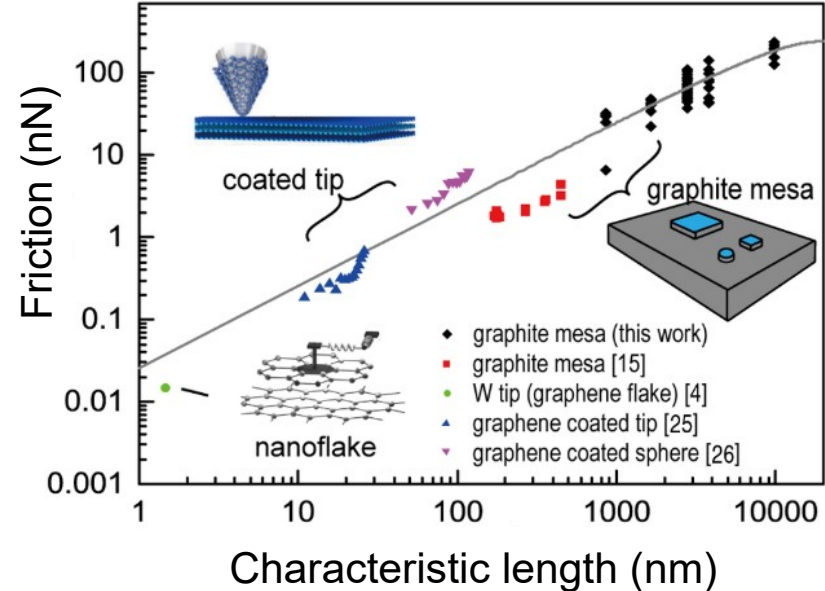
Edge characterizations

- Distorted layer
- sp^2 , sp^3
- O-groups



Scaling law of superlubricity

$$F_f = f_{\text{area}}L^2 + f_{\text{edge}}L$$



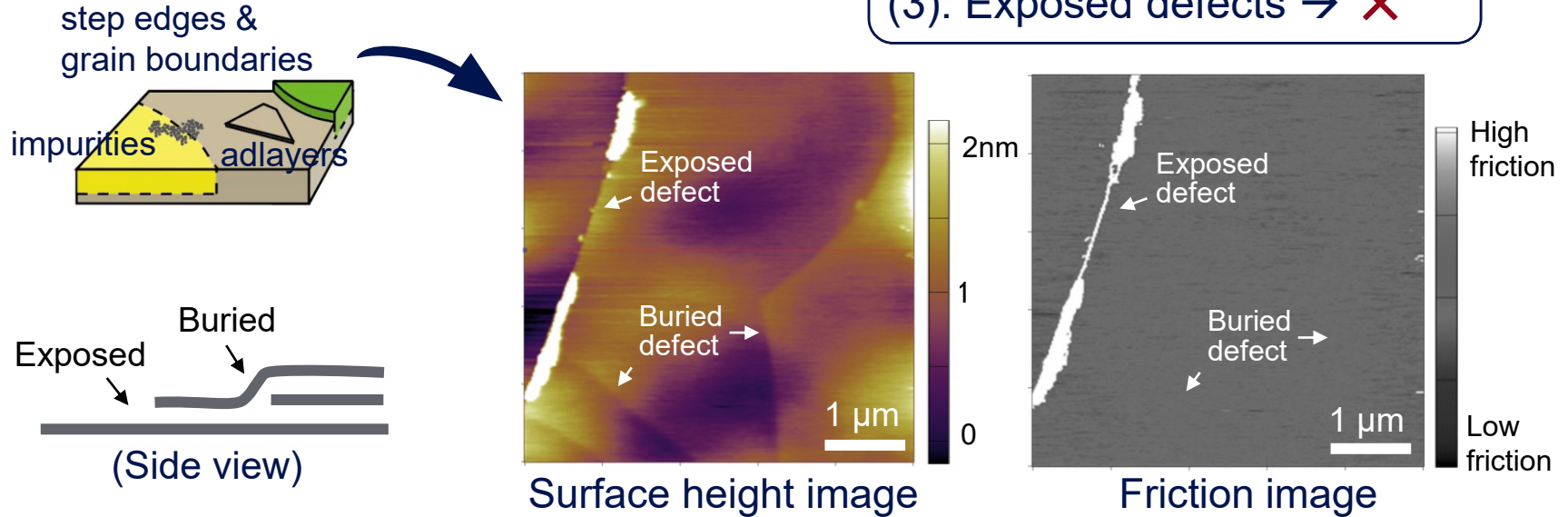
Defects Limit the Size of Superlubricity

Superlubricity fails for large contacts !

Current limit: $\sim 20\ \mu\text{m}$

Structure-property relation:

- (1). No defects $\rightarrow \checkmark$
- (2). Buried defects $\rightarrow \checkmark$
- (3). Exposed defects $\rightarrow \times$

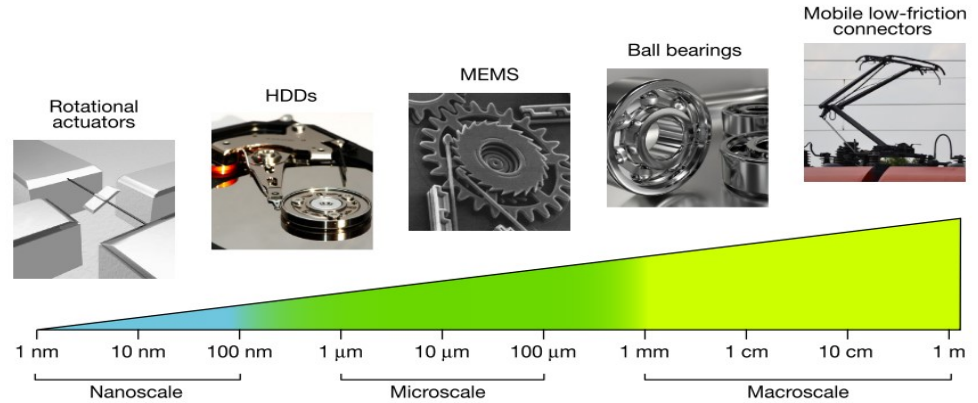


- K. Wang, C. Qu^{*}, J. Wang, B. Quan, and Q. Zheng^{*}, *Phys. Rev. Lett.* 125, 026101 (2020).
- D. Yang, C. Qu^{*}, Y. Gongyang, and Q. Zheng^{*}, *ACS Appl. Mater. Interfaces* 15, 44563 (2023).

Opportunities and Challenges

Scaling-up of superlubricity

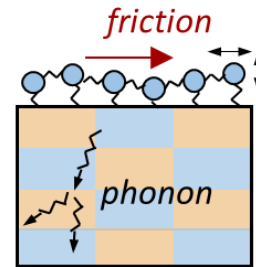
- Superlubric material fabrication
- Device development with current size constraints



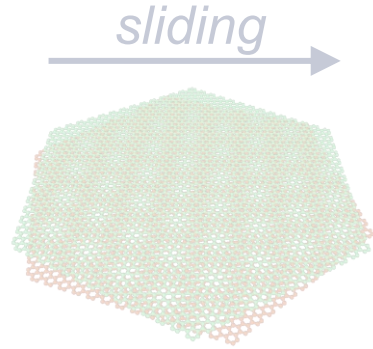
O. Hod, E. Meyer, Q. Zheng & M. Urbakh *Nature* 563, 485–492 (2018).

New friction control paradigms

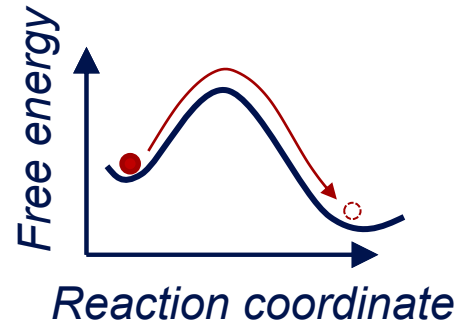
- Energy transport & dissipation
- Overlapping classical/quantum mechanics
- Relatively unexplored, potential for new discoveries



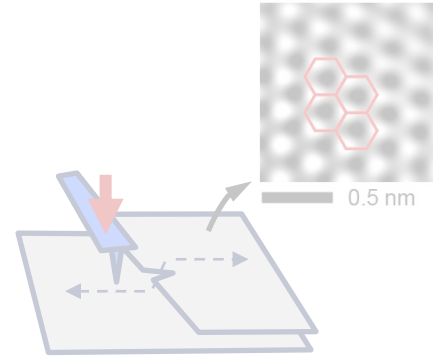
Outline



Structural
Superlubricity



Mechano-
chemistry



Nanofracture of
2D Materials

Mechanochemistry: Force-Driven Reactions



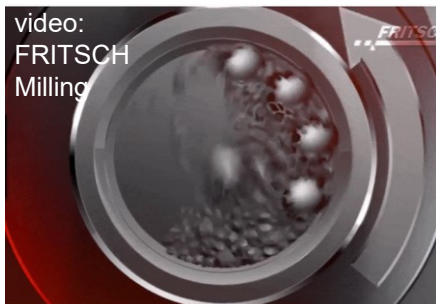
Mechanochemistry: chemical reactions modified by force and stress

- Green chemistry:
 - Environmentally friendly
 - High efficiency
- One of ten “emerging technologies in chemistry” (2019, IUPAC)

Chemical synthesis:

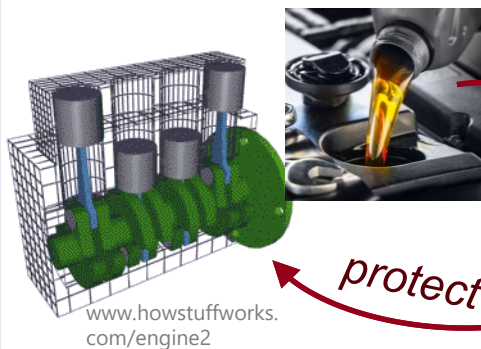


$\text{HgS} + \text{Cu} \rightarrow \text{Hg} + \text{CuS}$
~4th century B.C.

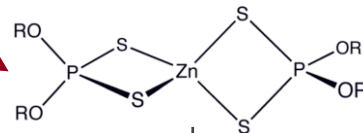


Modern reactor

Mechanical interfaces:



Zinc dialkyldithiophosphates
(ZDDP)



T, σ

**In-situ anti-wear
coating**

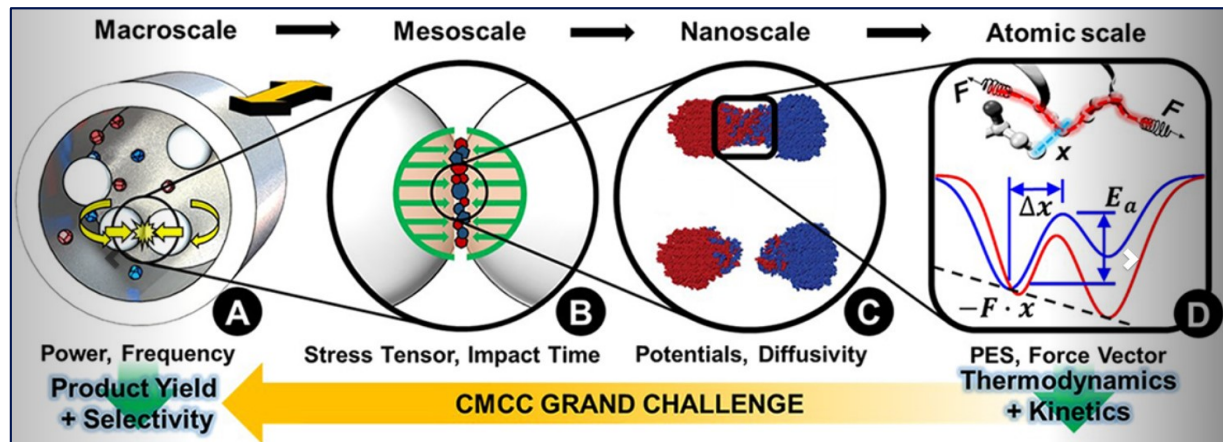
protect

Mechanochemistry: Force-Driven Reactions



Mechanochemistry: chemical reactions modified by force and stress

- Green chemistry:
 - Environmentally friendly
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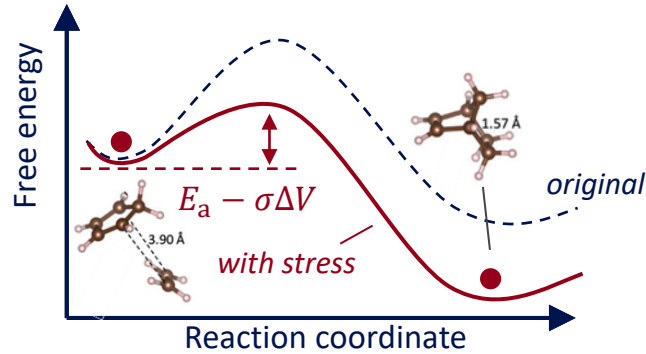
Example:

US NSF - Center for the Mechanical Control of Chemistry

- Collaboration: 10+ institutes
- \$20+ million total funding
- 2-phase project 2020-2028

Mechanism and Knowledge Gap

Theory: thermal activation

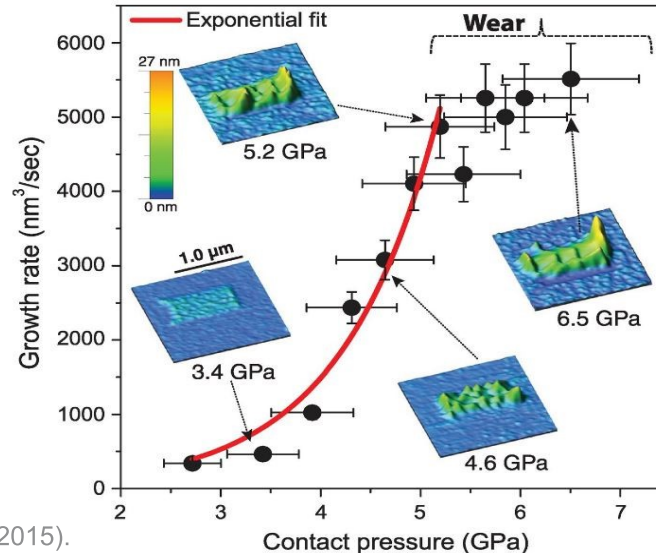
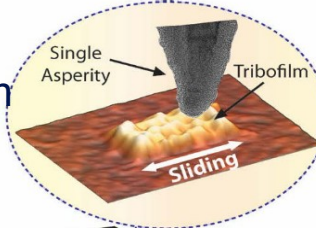


$$\Gamma = \Gamma_0 \exp\left(-\frac{E_a - \sigma\Delta V}{k_B T}\right)$$

Γ : Reaction rate
 Γ_0 : Pre-exponential factor
 E_a : Activation energy
 σ : Stress
 ΔV : Activation volume
 $k_B T$: Thermal energy

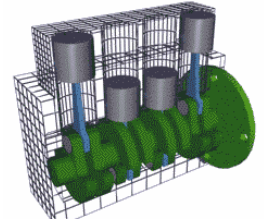
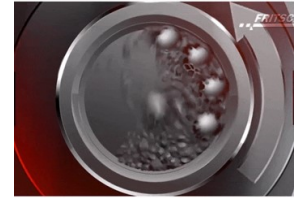
Nanoscale experiments

ZDDP film growth



Contact mechanics
Gap

Macroscale

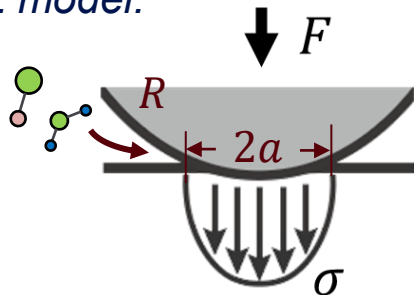


Engineering Interface:

- Geometry & deformation
- Complicated stress state
- Hard to measure/control

Adding Contact Mechanics to Mechanochemistry

Hertz model:



#1 Contact radius:

$$a = \frac{3R\pi}{4E^*} \sigma_m$$

#2 Stress distribution:

$$\sigma(r) = \frac{3}{2} \sigma_m \sqrt{1 - \left(\frac{r}{a}\right)^2}$$

R : ball radius

a : contact radius

σ_m : mean stress

E^* : reduced Young's modulus

ρ : reactant areal concentration

Γ : reaction rate

Γ_0 : pre-factor

$$\text{Reaction rate: } \Gamma = \iint_A \rho \Gamma_0 e^{-\frac{E_a - \sigma \Delta V}{k_B T}} dA$$

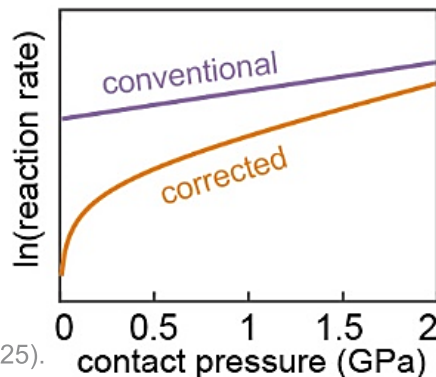
$$\rightarrow \Gamma = \int_0^a \rho \Gamma_0 e^{-\frac{E_a - \Delta V \sigma(r)}{k_B T}} 2\pi r dr$$

#1: Contact radius

$$(\lambda = \frac{3\Delta V}{2k_B T})$$

$$\rightarrow \Gamma = \frac{\pi^3}{2} \rho \Gamma_0 e^{-\frac{E_a}{k_B T}} \left(\frac{R k_B T}{E^* \Delta V}\right)^2 [1 + e^{\lambda \sigma_m} (\lambda \sigma_m - 1)]$$

The “contact-mechanics-corrected” model



Compare with conventional:

$$\Gamma = \Gamma_0 \exp\left(-\frac{E_a - \sigma_m \Delta V}{k_B T}\right)$$

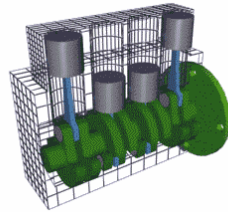
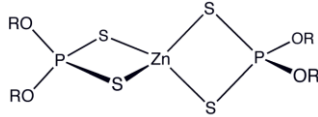
Reliable Measurement of ΔV : The Problem

Problem: literature values of activation volume scatter by a lot!

Examples:

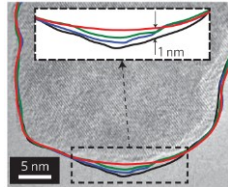
Growth of antiwear film from ZDDP

$$\Delta V = 3.8 \sim 180 \text{ \AA}^3$$

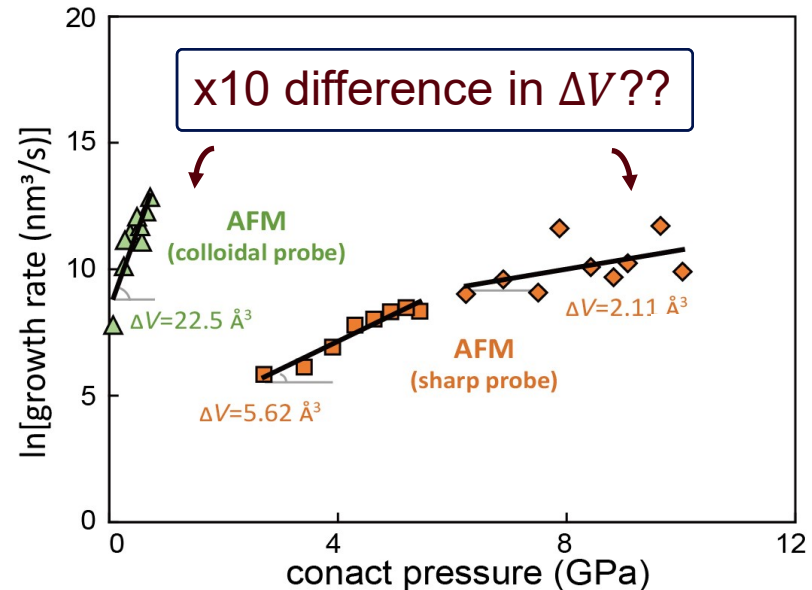


Tribochemical material removal (wear) of Si:

$$\Delta V = 6.7 \sim 60 \text{ \AA}^3$$



$$\ln \Gamma = \left[\frac{\Delta V}{k_B T} \right] \sigma + \ln \Gamma_0 - \frac{E_a}{k_B T}$$



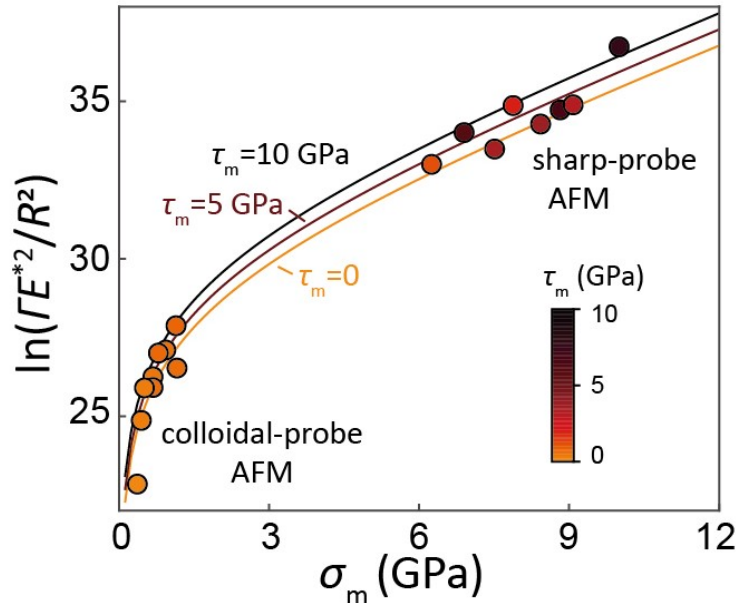
- A. Martini and S. H. Kim, *Tribol. Lett.* 69, 150 (2021).
- T. D. B. Jacobs and R. W. Carpick, *Nat. Nanotechnol.* 8, 108 (2013).

- C. Qu, L. Fang, and R. W. Carpick, *Phys. Rev. B* 111, 195405 (2025).

Reliable Measurement of ΔV : The Solution

The new “contact-mechanics-corrected” model:

$$\ln \left(\frac{\Gamma E^{*2}}{R^2} \right) = F(\sigma_m, \tau_m, T; \Delta V_n, \Delta V_s)$$



Unified values for activation volume:

Fitting to new model:

$$\begin{aligned} \Delta V_n &= -2.08 \pm 0.36 \text{ \AA}^3 \\ \Delta V_s &= 0.43 \pm 0.57 \text{ \AA}^3 \end{aligned} \rightarrow \Delta V_{\text{eff}} = \mu_m \Delta V_s - \Delta V_n = 2.3 \text{ \AA}^3$$

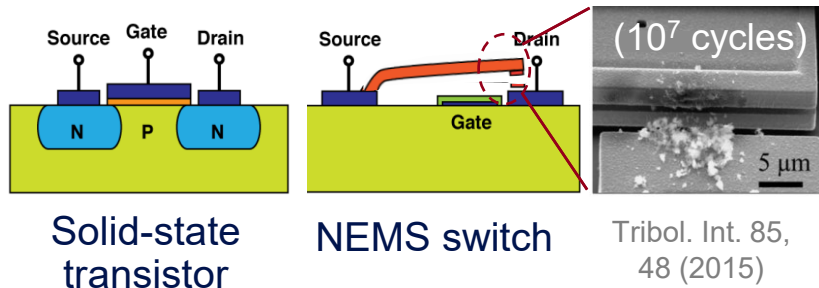
Previous results:

$$\Delta V_{\text{eff}} = 2.1 \text{ \AA}^3 \sim 22.5 \text{ \AA}^3$$

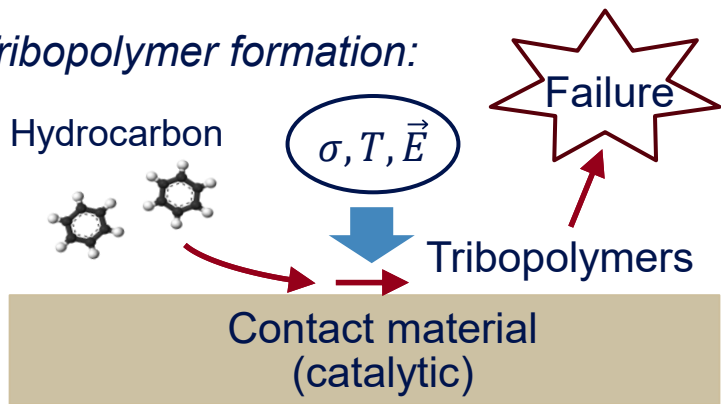
over-estimated!

Application in Nano-Electro-Mechanical Systems (NEMS)

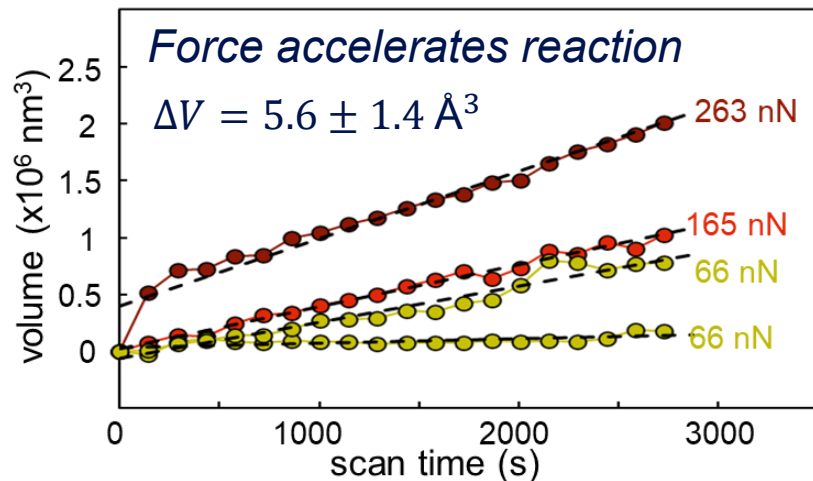
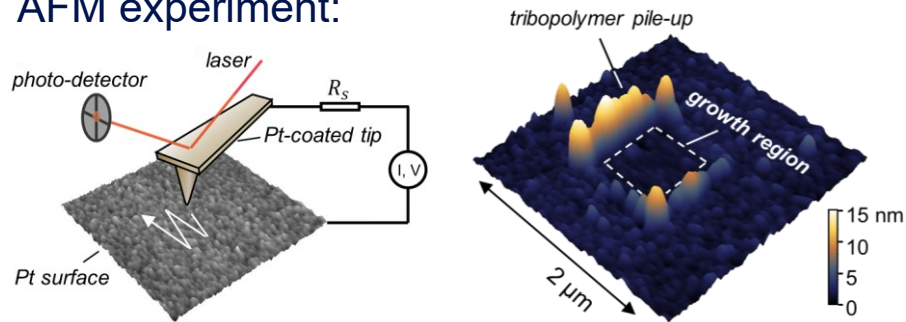
Background:



Tribopolymer formation:

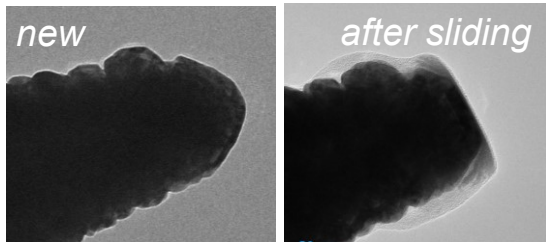


AFM experiment:

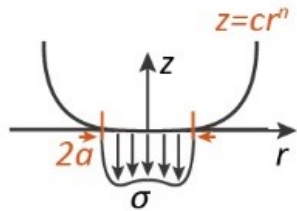


Application to Nano-Electro-Mechanical Systems (NEMS)

Extending the contact-mechanics-corrected model to flattened tips:



50 nm

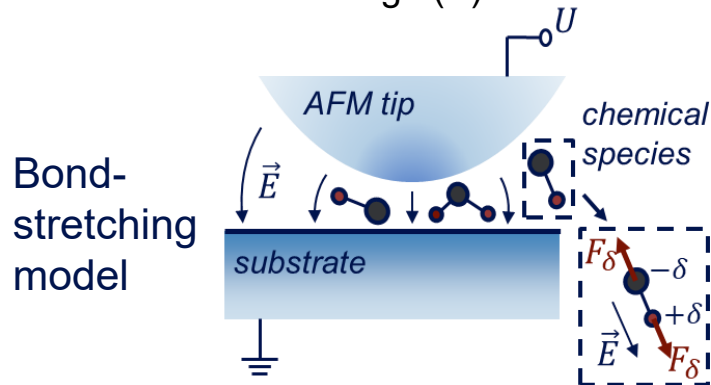
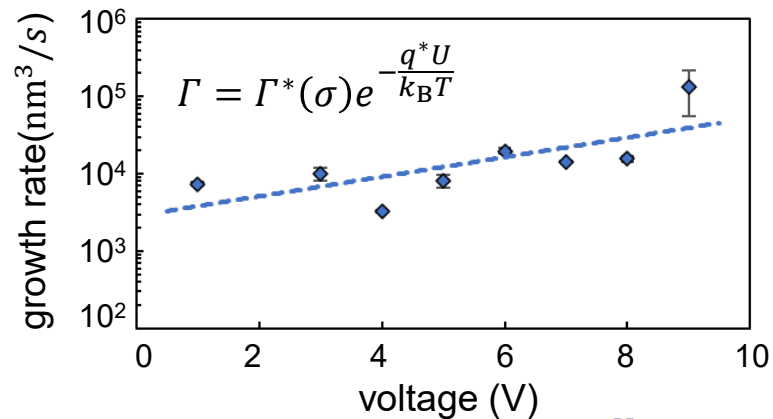


Power-law shaped tip:

$$z = cr^n$$

$$\Gamma = \underbrace{2\pi A_0 \left(\frac{\pi}{E^* \kappa(n) c} \cdot \frac{n+1}{2n} \right)^{\frac{2}{n-1}} e^{-\frac{E_a}{k_B T}} \cdot \sigma_m^{n-1}}_{\text{constant}} \underbrace{\int_0^1 e^{\frac{\Delta V^* \sigma_m \cdot \Sigma(n; \xi)}{k_B T}} \xi d\xi}_{\text{function of } \sigma_m}$$

Effect of voltage:



Opportunities and Challenges

The activation volume tensor:

Hydrostatic P :

(IUPAC definition, 1994)

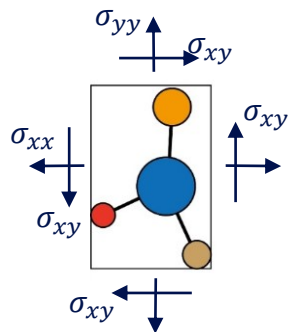
$$\Delta V = V_{\text{transition}} - V_{\text{initial}}$$

General σ_{ij} :

$$\frac{\Delta V_{ij}}{\uparrow} = \left(\frac{\partial \Delta G}{\partial \sigma_{ij}} \right)_T \sim \frac{\varepsilon_{ij}}{\uparrow}$$

experiment

simulation



Mechanochemistry of rough interface:

$$(\xi_{\sigma_0} = \frac{\sigma_0}{E^* \sqrt{m}}, \xi_{\sigma_Y} = \frac{\sigma_Y}{E^* \sqrt{m}}, \xi_{\Delta V} = \frac{\Delta V E^* \sqrt{m}}{k_B T})$$

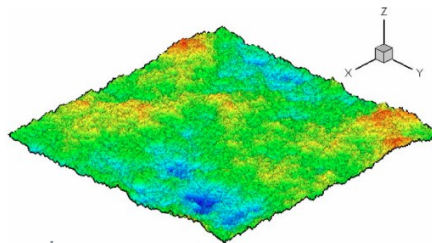


Image:

S. Hyun, et al, *Phys. Rev. E* 70, 12 (2004)

$$\text{Plastic: } \Gamma \sim \frac{\xi_{\sigma_0}}{\xi_{\sigma_Y}} \cdot e^{\xi_{\Delta V} \xi_{\sigma_Y}}$$

$$\text{Elastic: } \Gamma \sim \frac{2}{\pi} \xi_{\sigma_0} h(\xi_{\Delta V})$$

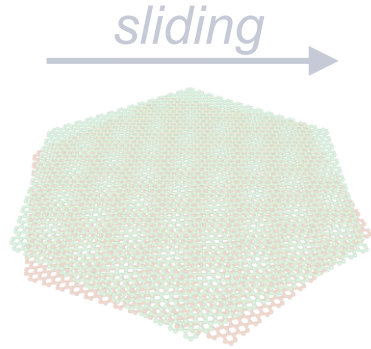
$$\Gamma \sim e^{\xi_{\Delta V} \xi_{\sigma_0}}$$

C. Qu, R. W. Carpick (in preparation)

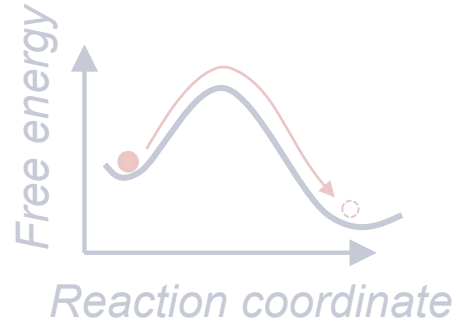
Goal:

- To make mechanochemistry understandable and predictable
- New methods, new designs, better performance

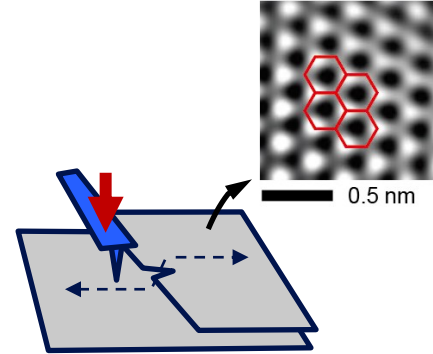
Outline



Structural
Superlubricity



Mechano-
chemistry



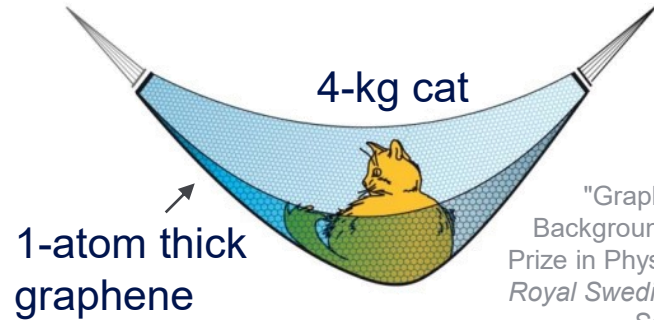
Nanofracture of
2D Materials

Nanofracture of 2D materials

Mechanical properties of graphene:

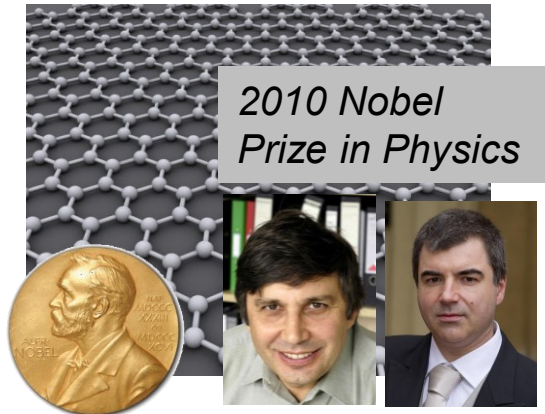
Young's modulus: ~ 1 TPa ($>5\times$ steel)

Strength: ~ 130 GPa ($>100\times$ steel)

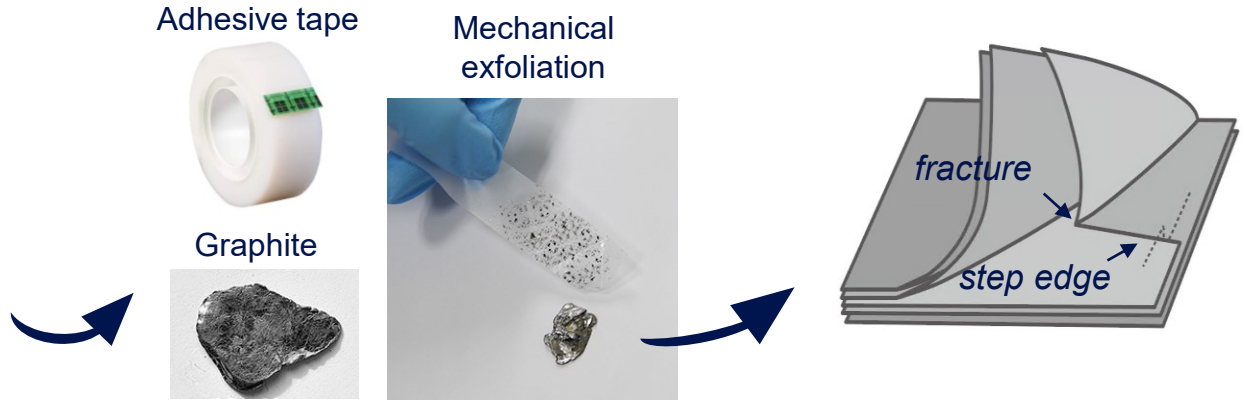


"Graphene: Scientific Background on the Nobel Prize in Physics 2010." *The Royal Swedish Academy of Sciences* (2010).

Mechanical exfoliation to fabricate graphene:

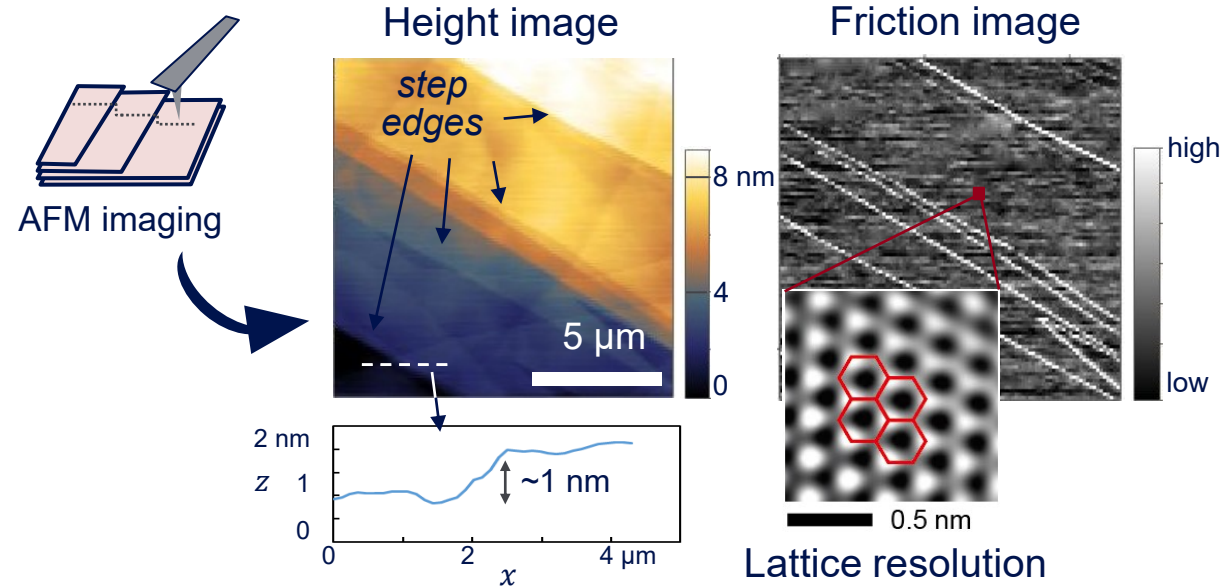


A. Geim & K. Novoselov



A New AFM Approach

Step edges as imaged by atomic force microscopy (AFM):

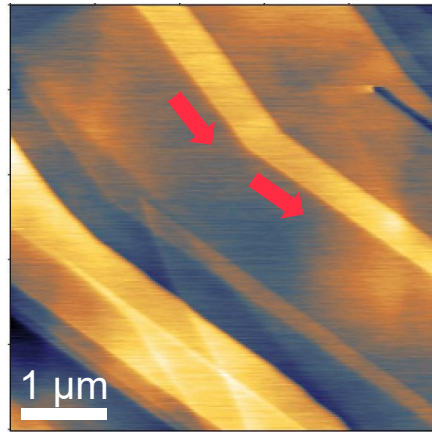


A new AFM approach:

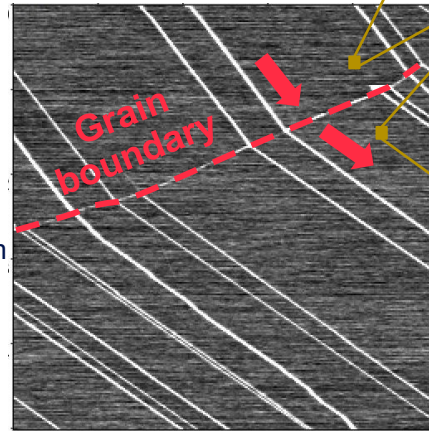
- High-throughput
- High spatial resolution (sub-nanometer)
- Combining mechanics models

Study 1: Anisotropic Fracture of Graphene

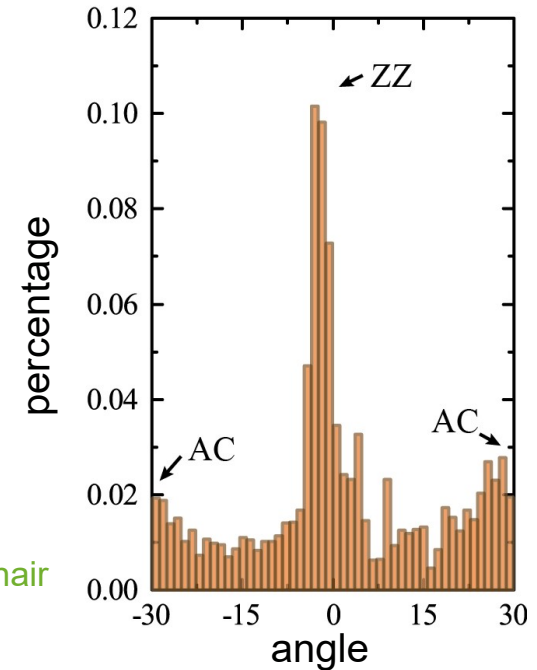
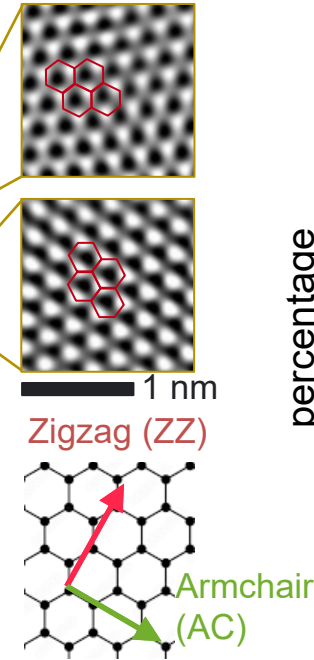
*Step edges as imaged by AFM:
They follow a certain lattice orientation!*



AFM height image



AFM friction image

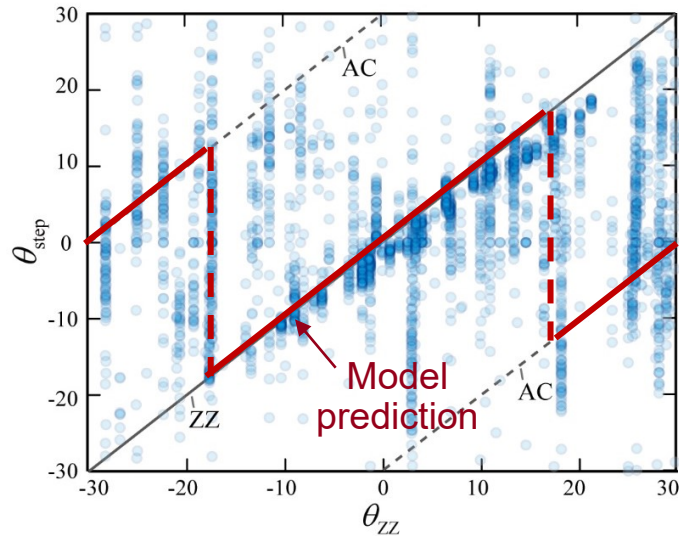
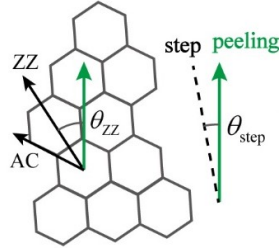
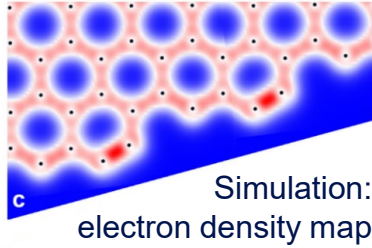


This means:

The in-plane fracture of graphene is anisotropic

Large dataset:
~100 grains,
~3000 step edges

Mechanics Modelling



Anisotropic fracture toughness:

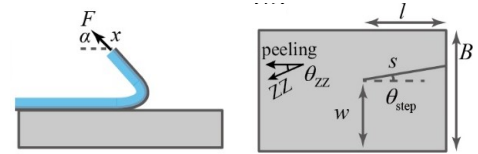
$$G_c(\theta) = 2\gamma(\theta) \sim \gamma_{AC} \sin \theta + \gamma_{ZZ} \sin(30^\circ - \theta)$$

Y. Liu, A. Dobrinsky, and B. I. Yakobson, *Phys. Rev. Lett.* 105, 235502 (2010).

Energy analysis:

$$U = U_E + 2\gamma ts/d + \tau Bl$$

Elastic energy Edge energy Delamination energy



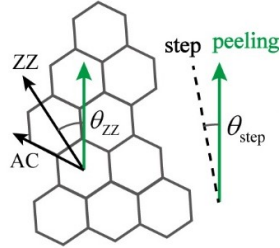
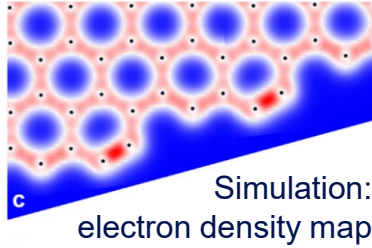
Equilibrium crack extension & max energy release rate:

$\partial U / \partial s = 0, \dots \rightarrow$ Criterion for crack direction:

$$\theta_{peel} - \theta_{zz} \geq \tan^{-1} \left(2 \frac{\gamma_{AC}}{\gamma_{ZZ}} - \sqrt{3} \right)$$

External force Lattice orientation Fracture anisotropy

Mechanics Modelling



Anisotropic fracture toughness:

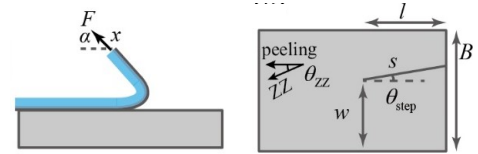
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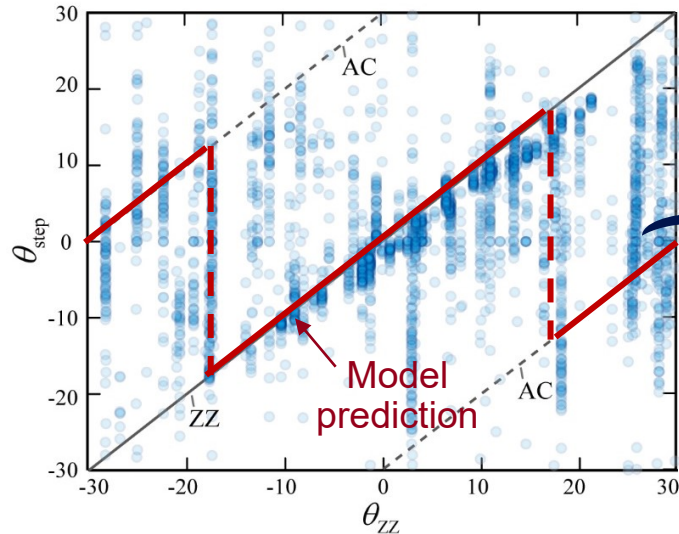


Equilibrium crack extension & max energy release rate:

$$\partial U / \partial s = 0 \Rightarrow \dots$$

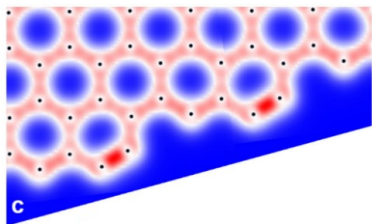
A measurement on graphene's fracture toughness anisotropic factor

$$\frac{G_{ZZ}}{G_{AC}} = \frac{\gamma_{ZZ}}{\gamma_{AC}} = 0.971$$



Implications of the Anisotropic Fracture

Edge atom density & energy



$$\frac{\rho_{ZZ}}{\rho_{AC}} = \frac{\sqrt{3}}{2} \approx 0.866$$

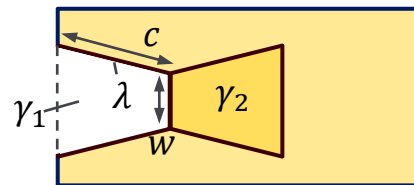
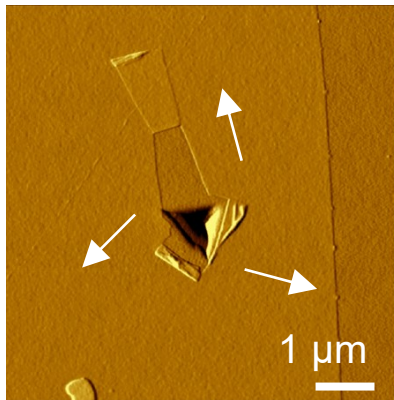
$$\frac{\gamma_{ZZ}}{\gamma_{AC}} = 0.971$$

Benchmark for simulations

Method	γ_{ZZ}/γ_{AC}
Experiment (present work)	0.971
Density functional theory	0.95~1.34
Molecular dynamics	0.69~1.05

- C. Qu, et al. *Phys. Rev. Lett.* 129, 026101 (2022).
- H. Yin, et al. *Nano Lett.* 15, 1918 (2015).
- C. K. Gan and D. J. Srolovitz. *Phys. Rev. B* 81, 125445 (2010).
- Y. Liu, A. Dobrinsky, and B. I. Yakobson. *Phys. Rev. Lett.* 105, 235502 (2010).
- P. Zhang et al. *Nat. Commun.* 5, 3782 (2014).

Spontaneous self-assembly of graphene



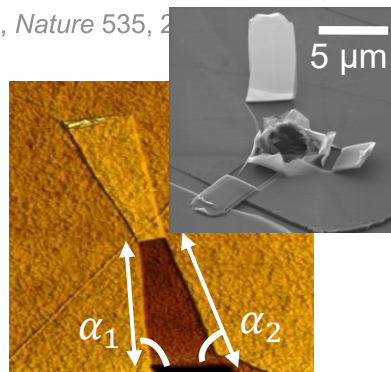
$$(\gamma_2 - \gamma_1) \frac{w \delta l}{\delta c} \geq \frac{\delta U_{\text{fold}}}{\delta c} + 2\lambda$$

Surface energy
Elastic energy
Fracture energy

- J. Annett and G. L. W. Cross, *Nature* 535, 201 (2016).

- Anisotropic fracture energy $\lambda(\alpha)$
- Controlled self-cutting

L. Capaldi, L. Yuan, C. Qu, D. Sanchez, R. Carpick, O. Tertuliano, (under review in *Nano Lett.*)



Study 2: Detecting Fracture Initiation with AFM

Fracture initiation in step edges:

1. Reciprocal scanning
at a fixed location
(cutting attempts)

2. Fracture event detected
($N=198$)

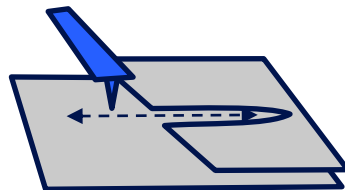
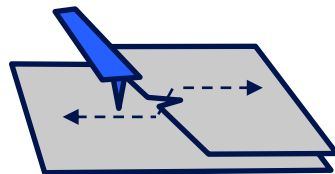
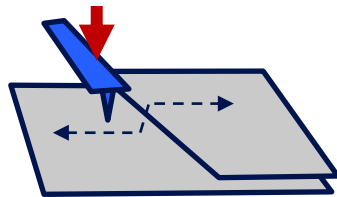
fracture initiation rate:

$$f \stackrel{\text{def}}{=} 1/t$$

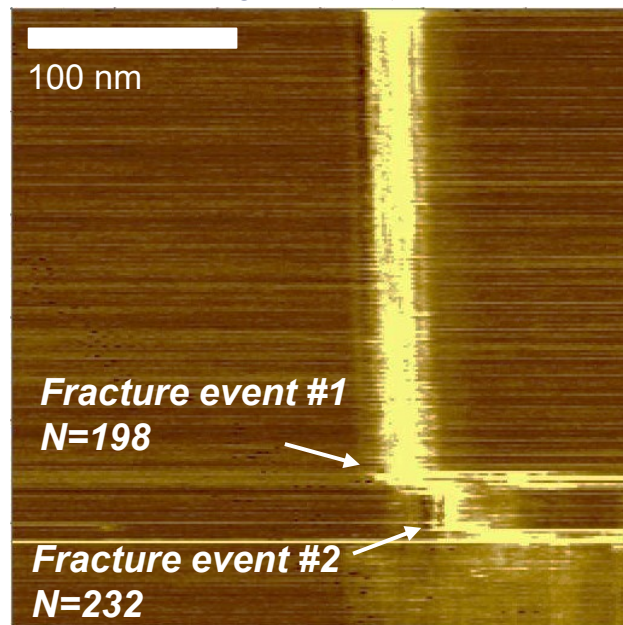
(t : time exposed to stress)

3. Further fracture events

normal force



Friction image



Step edge
(showing high friction)

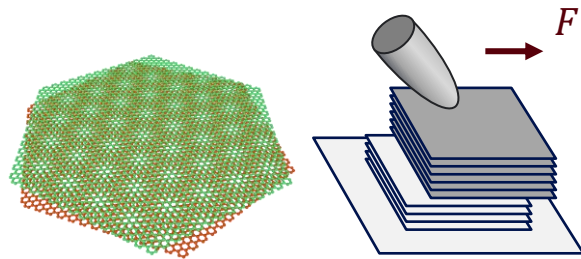
$N=1$ Time sequence $N=256$

friction
high
low

Summary

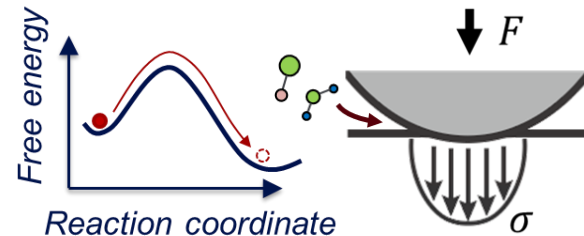
Structural superlubricity

- ❑ Achieving “zero friction” without lubricants
- ❑ Unique behaviors at “zero friction”
- ❑ Mechanisms & limitations (edge, defects, size)



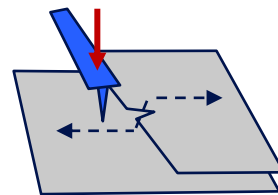
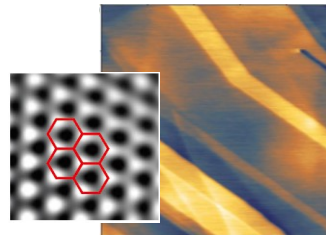
Mechanochemistry

- ❑ Green synthesis & engineering interfaces
- ❑ New insight: contact mechanics + mechanochemistry



Nanofracture of 2D Materials

- ❑ A new AFM-based experimental approach
- ❑ Fracture anisotropy ❑ Fracture initiation



Acknowledgements



Zheng Group @ Tsinghua



PhD supervisor:
Prof. Quanshui
Zheng



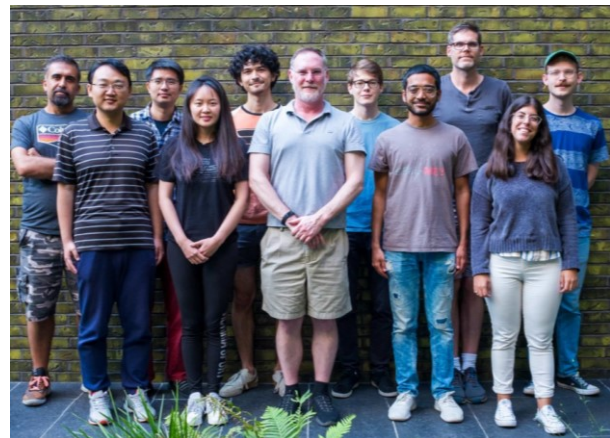
Prof. Ming
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(Tsinghua)



Prof.
Zhiping Xu
(Tsinghua)



Prof. M.
Urbakh (Tel
Aviv Univ.)



Carpick Group @ UPenn



Prof. R. Carpick
(UPenn)



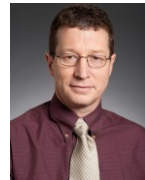
Prof. D.
Srolovitz



Prof. G. Cross
(TCD Ireland)



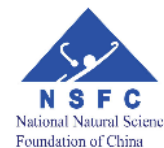
Prof. A. Rappe
(UPenn)



Prof. M. de
Boer (CMU)



Prof. G.
Piazza (CMU)



Thank you !!

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